

# Fabrication and Photovoltaic Properties of Heterostructured TiO<sub>2</sub> Nanowires

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One-dimensional heterostructured TiO<sub>2</sub> nanowires were successfully fabricated by an electrospinning technique and modified by hydrolysis. We investigated their structure, morphology, chemical composition, and optical properties by using the X-ray diffraction (XRD), scanning electron microscopy (SEM), transmission electron microscopy (TEM), X-ray photoelectron spectroscopy (XPS), and UV-vis spectroscopy. In the case of the photovoltaic performance, the short-circuit current density and cell efficiency of the DSSCs employing single TiO<sub>2</sub> nanowires and heterostructured TiO<sub>2</sub> nanowires improve from 6.90 to 11.38 mA/cm<sup>2</sup> and from 2.56 to 4.29%, respectively. The results show that the photoconversion efficiency of the heterostructured TiO<sub>2</sub> nanowires could be improved by more than ~67% compared to that of the single TiO<sub>2</sub> nanowires because of the enhanced specific surface area that facilitates dye adsorption.

**Keywords:** TiO<sub>2</sub>, Heterostructured Nanowires, Electrospinning, Photovoltaic Properties.

## 1. INTRODUCTION

Dye-sensitized solar cells (DSSCs) are presently receiving significant attention as potential alternatives to conventional crystalline solar cells because of their high conversion efficiency and low manufacturing cost.<sup>1</sup> A DSSC consists of an *n*-type semiconductor electrode such as TiO<sub>2</sub> and ZnO with a solar-energy absorbing dye, an electrolyte, and a counter electrode; the cell is a photovoltaic device based on the principles of plant photosynthesis.<sup>1–3</sup>

Titanium dioxide (TiO<sub>2</sub>) is an *n*-type semiconductor that has wide applications as a photocatalyst, catalyst support, dielectric material, photovoltaic electrode, and pigment materials.<sup>3–5</sup> In particular, among the various applications, TiO<sub>2</sub> has been widely used as photoelectrodes in photovoltaic cells such as DSSCs. Considerable effort has been devoted to improve the performance of DSSCs via various synthetic methods such as sol-gel, hydrolysis, hydro-/solve-thermal, anodizing, vacuum deposition, sonochemical, and microwave-assisted methods. In addition, one-dimensional TiO<sub>2</sub> nanostructures have been actively studied because of their unusual and unique characteristics.<sup>4–7</sup> For example, Kang et al. reported the fabrication of highly ordered TiO<sub>2</sub> nanotubes with an

efficiency of 3.5% using a nanoporous alumina templating method.<sup>8</sup> Moreover, Feng et al. presented tantalum-doped TiO<sub>2</sub> nanowire arrays hydrothermally synthesized for DSSCs with a conversion efficiency of 4.1%.<sup>9</sup> Oh et al. argued that TiO<sub>2</sub>-branched nanostructure electrodes that were fabricated by a seeding method had a conversion efficiency of 4.3%.<sup>10</sup> The study of the use of one-dimensional nanostructures in DSSCs is underway, and the relation between the one-dimensional nanostructures and the photovoltaic performances has not been clearly understood hitherto.

In this study, heterostructured TiO<sub>2</sub> nanowire (NW) electrodes, consisting of TiO<sub>2</sub> nanoparticles decorated on TiO<sub>2</sub> nanowires, were successfully fabricated by means of an electrospinning method with a hydrolysis reaction. Subsequently, their structural properties and photovoltaic performances of DSSCs were investigated.

## 2. EXPERIMENTAL DETAILS

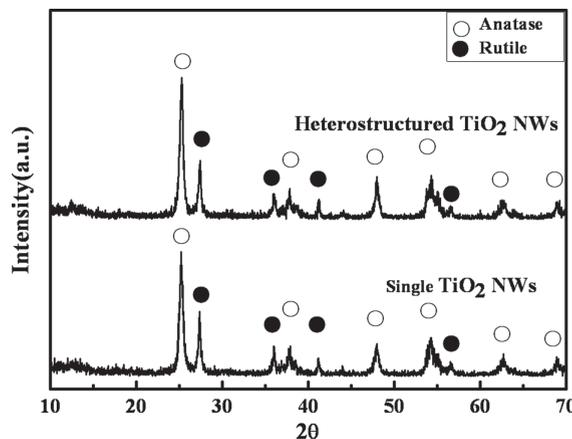
TiO<sub>2</sub> NWs were synthesized using an electrospinning method. In the electrospinning process, titanium tetraisopropoxide (Ti(O*i*Pr)<sub>4</sub>, Aldrich) was mixed with anhydrous ethanol and acetic acid. After stirring for 10 min, the solution was mixed with the polymer carrier solution of poly(vinylpyrrolidone) (PVP, *M<sub>w</sub>* = 1,300,000 g/mol,

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Aldrich) dissolved in ethanol, and then stirred for 30 min. Then, the blended solution was placed in a syringe with a 23-gauge stainless steel needle. The feeding rate and the tip-to-collector distance were controlled to be  $\sim 0.04$  ml/h and  $\sim 8$  cm, respectively, under an applied voltage of 8 kV during electrospinning. Single TiO<sub>2</sub> nanowires were obtained by removing PVP from the as-spun nanowires via calcination at 500 °C for 5 h in air. In order to fabricate the heterostructured TiO<sub>2</sub> nanowires, electro-spun TiO<sub>2</sub> nanowires were dispersed in DI water, which was added to a precursor of Ti(OiPr)<sub>4</sub> mixed with 2-propanol solution. After hydrolysis for 6 h at room temperature, the heterostructured TiO<sub>2</sub> nanowires were washed by DI water. Then, heterostructured TiO<sub>2</sub> nanowires were successfully obtained by freeze-drying at  $-50$  °C. For the assembly of DSSCs, fluorine-doped tin oxide coated glasses (FTO, 15  $\Omega/\text{cm}^2$ , Solaronix) were ultrasonically cleaned by acetone, methanol, ethanol, and DI water. The DSSCs consisted of a working electrode, a counter electrode, and an electrolyte, respectively. For the fabrication of the working electrodes, a TiO<sub>2</sub> paste was prepared by using 16.2 wt% of single/heterostructured TiO<sub>2</sub> nanowires, 11.9 wt% of hydroxypropyl cellulose (HPC,  $M_w \sim 80,000$  g/mol, Aldrich), 4.4 wt% of acetyl acetone (Aldrich), and 67.5 wt% of DI water. The mixture was ground in an agate mortar for 6 h to obtain a viscous paste. Subsequently, the single/heterostructured TiO<sub>2</sub> films were fabricated by coating the prepared TiO<sub>2</sub> paste on the cleaned FTO glass using a squeeze printing technique and then sintering at 500 °C for 1 h. The calcined TiO<sub>2</sub> electrodes were immersed in 0.5 mM Ru(dcbpy)<sub>2</sub>(NCS)<sub>2</sub> (535-bis TBA, Solaronix) and dissolved in ethanol overnight ( $\sim 24$  h). The Pt-coated FTO glass counter electrode was prepared by coating with platinum paste (PT-1, Dyesol) and sequential sintering at 450 °C for 30 min. The prepared working and counter electrodes were assembled into a sandwich, and filled with a 0.3 M DMPII based electrolyte.<sup>11</sup> The structural properties of the heterostructured TiO<sub>2</sub> NWs were established by X-ray diffraction (XRD, Rigaku X-ray diffractometer equipped with a Cu K $\alpha$  radiation), scanning electron microscopy (SEM, Hitachi S-4700), and high-resolution electron microscopy (HREM, JEOL, 2100F, KBSI Suncheon center) analyses. The chemical bonding states of the samples were examined using X-ray photoelectron spectroscopy (XPS, ESCALAB 250 equipped with an Al K $\alpha$  X-ray source). The optical properties of the samples were investigated by using UV-vis spectrometer (Jasco V650). The photovoltaic properties of the DSSCs were evaluated by using a 150 W xenon lamp (LAB 50) with the standard irradiance level (100 mW cm<sup>-2</sup>) for the active sample areas of 0.196 cm<sup>2</sup>.

### 3. RESULTS AND DISCUSSION

We used X-ray diffraction (XRD) to investigate the structural properties of the single TiO<sub>2</sub> NWs and the

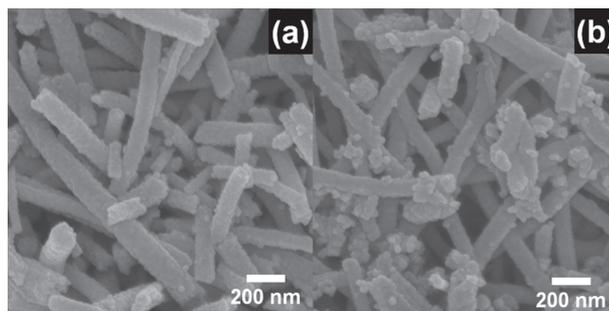


**Fig. 1.** XRD plots of the single TiO<sub>2</sub> nanowires and the heterostructured TiO<sub>2</sub> nanowires.

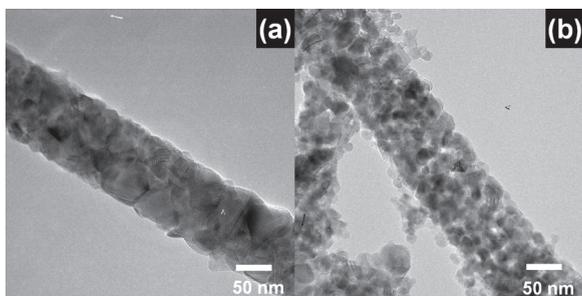
heterostructured TiO<sub>2</sub> NWs, as shown in Figure 1. The single TiO<sub>2</sub> NWs are well-crystallized via calcination at 500 °C, as indicated by the sharp diffraction peaks for the anatase/rutile mixed structure. The characteristic diffraction peaks at 25.3°, 37.8°, 48.0°, 53.9°, 62.7°, and 68.8° correspond to the (101), (004), (200), (105), (204), and (116) planes, respectively, of anatase TiO<sub>2</sub> (JCPDS No. 841286). In addition, the peaks at 27.9°, 36.4°, 41.7°, and 55.1° correspond to the (110), (101), (111), and (211) planes of rutile TiO<sub>2</sub> (JCPDS No. 881175). For the characteristic peaks of the single TiO<sub>2</sub> NWs, the phase composition of the TiO<sub>2</sub> nanocomposite can be estimated according to the equation below:

$$W_R = A_R / (0.884 \cdot A_A + A_R)$$

where  $W_R$ ,  $A_A$ , and  $A_R$  are the weight percentage of the rutile structure, the integrated intensity of the (101) anatase peak, and the integrated intensity of the (110) rutile peak, respectively.<sup>12</sup> The rutile weight percentage in the single TiO<sub>2</sub> NWs was estimated to be  $\sim 27.4\%$ , whereas the heterostructured TiO<sub>2</sub> NWs have a relatively anatase-rich crystal structure with  $\sim 22\%$  rutile content compared to the single TiO<sub>2</sub> NWs. This result indicates that the increase of anatase content on the NW surface may be attributed to the NW modification via the hydrolysis reaction in titanium



**Fig. 2.** SEM images of (a) the single TiO<sub>2</sub> nanowires and (b) the heterostructured TiO<sub>2</sub> nanowires.

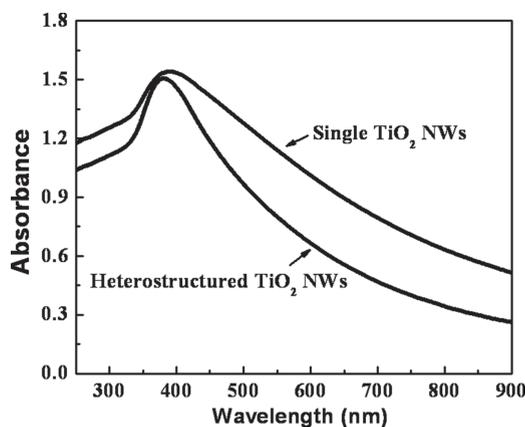


**Fig. 3.** HRTEM images obtained from (a) the single TiO<sub>2</sub> nanowires and (b) the heterostructured TiO<sub>2</sub> nanowires.

tetraisopropoxide (TTIP) solution. It has been known that anatase TiO<sub>2</sub>-based solar cells exhibit better photoelectrochemical performances compared to rutile TiO<sub>2</sub>-based solar cells owing to the large surface area that facilitates dye adsorption. This property can be easily confirmed from the photocurrent density–voltage curves.<sup>13</sup>

Figure 2 shows SEM images of the single TiO<sub>2</sub> NWs and the heterostructured TiO<sub>2</sub> NWs. The single TiO<sub>2</sub> NWs have a smooth surface with a diameter of ~80–130 nm after calcination at 500 °C, and the nanoparticles that formed on the nanowire surface of the heterostructured TiO<sub>2</sub> NWs via the hydrolysis of the TTIP precursor as reported elsewhere<sup>14,15</sup> had formed with the size of several nm. The rutile/anatase ratio change mentioned in Figure 1 is probably caused by the formation of such a second phase. However, the diameter of the support nanowires remained the same as that of the single NWs after the TTIP treatment, and the nanoparticles were randomly distributed on the TiO<sub>2</sub> NW surface, which may increase the number of adsorption sites for dye molecules.

Figure 3 shows the HRTEM images obtained from the single and the heterostructured TiO<sub>2</sub> NWs. The single TiO<sub>2</sub> NW shows a dense crystal structure with a diameter of ~100 nm after calcination at 500 °C. However, the heterostructured TiO<sub>2</sub> NW obtained by post TTIP treatment of electro-spun TiO<sub>2</sub> NW seems to consist of crystalline nanoparticles with a size of 5–10 nm covering the dense NW surface. In other words, smaller grains form on

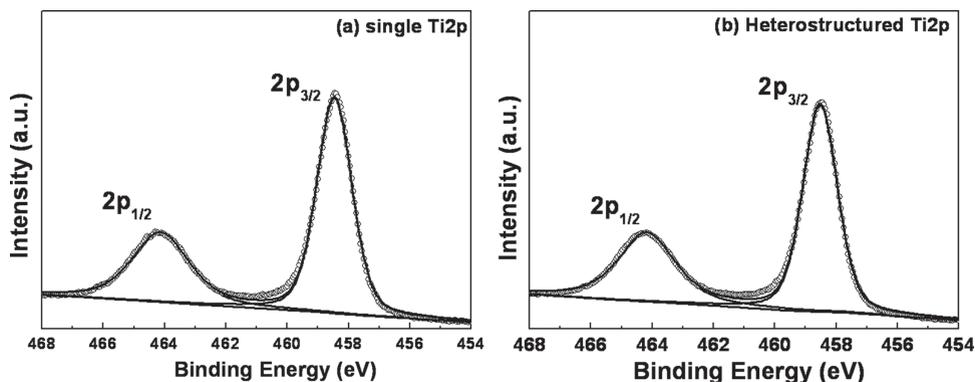


**Fig. 5.** UV-vis absorption spectra of the single TiO<sub>2</sub> nanowires and the heterostructured TiO<sub>2</sub> nanowires.

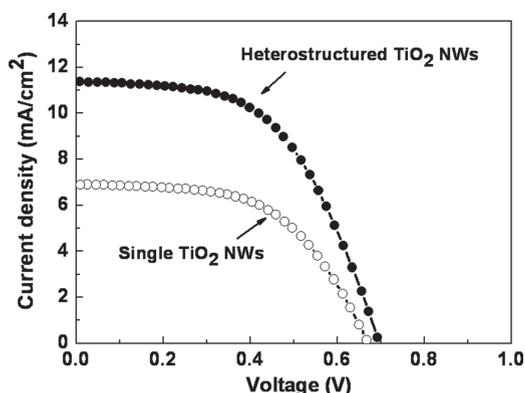
the heterostructured TiO<sub>2</sub> NWs relative to the single TiO<sub>2</sub> NWs. The heterostructure formation of the nanoparticles on NWs should contribute to an increase in the surface area. Therefore, such morphology modification of the TiO<sub>2</sub> NW surface will affect photovoltaic characteristics such as photocurrent density and conversion efficiency.

We conducted XPS measurements to examine the chemical bonding state of the samples. The photoelectron energies of the Ti 2*p* peaks were corrected by considering the reference C 1*s* peak at 284.5 eV. Figures 4(a) and (b) show the XPS core-level spectra for the Ti 2*p*<sub>3/2</sub> and Ti 2*p*<sub>1/2</sub> photoelectrons with binding energies of ~458.5 eV and ~464.2 eV for the single TiO<sub>2</sub> NWs and the heterostructured TiO<sub>2</sub> NWs, respectively. The characteristic peaks of the heterostructured TiO<sub>2</sub> NWs exhibit no noticeable difference relative to those of the single TiO<sub>2</sub> NWs. This implies that the elemental Ti in TiO<sub>2</sub> is not Ti(II) species but Ti(IV) species.<sup>16</sup>

Figure 5 displays the UV-vis absorption spectra for the single TiO<sub>2</sub> NWs and the heterostructured TiO<sub>2</sub> NWs. The TTIP treatment obviously affects light absorption characteristics of TiO<sub>2</sub> NWs as shown in Figure 5. Generally, the main absorption of pure anatase TiO<sub>2</sub> appears at wavelengths shorter than 400 nm, which corresponds to



**Fig. 4.** XPS core-level spectra of the Ti 2*p* photoelectrons for (a) the single TiO<sub>2</sub> nanowires and (b) the heterostructured TiO<sub>2</sub> nanowires.



**Fig. 6.** Photocurrent density–voltage curves measured from the DSSCs using the single TiO<sub>2</sub> nanowires and the heterostructured TiO<sub>2</sub> nanowires.

**Table I.** Photovoltaic performances of DSSCs employing the single TiO<sub>2</sub> NWs and heterostructured TiO<sub>2</sub> NWs.

Samples	$J_{sc}$ (mA/cm <sup>2</sup> )	$V_{oc}$ (V)	FF	$\eta$ (%)
Heterostructured TiO <sub>2</sub> NWs	11.38	0.69	0.54	4.29
Single TiO <sub>2</sub> NWs	6.90	0.67	0.55	2.56

its intrinsic bandgap of 3.2 eV.<sup>17</sup> The maximum absorption peak and the absorption edge of the heterostructured TiO<sub>2</sub> NWs show a blue shift (ca. 10 nm) in the bandgap transition compared to those of the single TiO<sub>2</sub> NWs. That is, the optical bandgap of the heterostructured TiO<sub>2</sub> NWs becomes larger slightly than that of the single TiO<sub>2</sub> NWs. This implies that the heterostructured TiO<sub>2</sub> NWs are anatase-rich phases. This result is in good agreement with XRD results as shown in Figure 1.

Two different types of TiO<sub>2</sub> NWs were used as the photoanode in the dye-sensitized solar cells (DSSCs). The photocurrent density ( $J$ )–voltage ( $V$ ) curves of the DSSCs assembled with single and heterostructured TiO<sub>2</sub> NWs are presented in Figure 6 and the derived DSSC curve parameters are summarized in Table I. For the case of DSSCs fabricated with single and heterostructured TiO<sub>2</sub> NWs, the  $V_{oc}$  changes are negligible. However, the DSSC fabricated with heterostructured TiO<sub>2</sub> NWs exhibits superior photovoltaic efficiency with enhanced photocurrent density compared to the DSSC fabricated with single TiO<sub>2</sub> NWs. In particular, the photoconversion efficiency (PCE;  $\eta$ ) of DSSCs with single TiO<sub>2</sub> NWs and heterostructured TiO<sub>2</sub> NWs increases significantly from 2.56% to 4.29%. This implies that the PCE of the heterostructured TiO<sub>2</sub> NWs is  $\sim 67\%$  greater than that of the single TiO<sub>2</sub> NWs. This result indicates that the higher PCE can be attributed to the improvement of the photocurrent density resulting from an enhanced specific surface area. Therefore, the morphology modification of such one-dimensional heterostructure by the hydrolysis reaction in TTIP solution may significantly enhance the

photoconversion efficiency of one-dimensional TiO<sub>2</sub> photoelectrodes rendering them useful for DSSCs.

#### 4. SUMMARY

We synthesized heterostructured TiO<sub>2</sub> NWs by nanoparticle formation via hydrolysis of the TTIP precursor solution on electro-spun TiO<sub>2</sub> NWs for DSSCs. SEM and TEM images showed that after calcination, the surface morphology of the TTIP-treated TiO<sub>2</sub> NWs becomes rougher surface than that of single TiO<sub>2</sub> NWs. The DSSC device with the surface-modified TiO<sub>2</sub> heterostructure also shows higher photovoltaic performance because of the enhancement of the photocurrent density as well as the improvement of light absorption via higher dye loading.

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